

Collection of Selected Presentation Materials from Recent Talks

Spintronic Devices for Nonvolatile VLSIs

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Collaborators: T. Endoh, T. Hanyu, H. Sato, S. Fukami, S. Kanai, S. Ikeda, F. Matsukura and the CSIS team

Work supported by the FIRST Program from JSPS
and partly by R & D for Next-Generation Information Technology of MEXT



<http://www.csis.tohoku.ac.jp/>



Outline



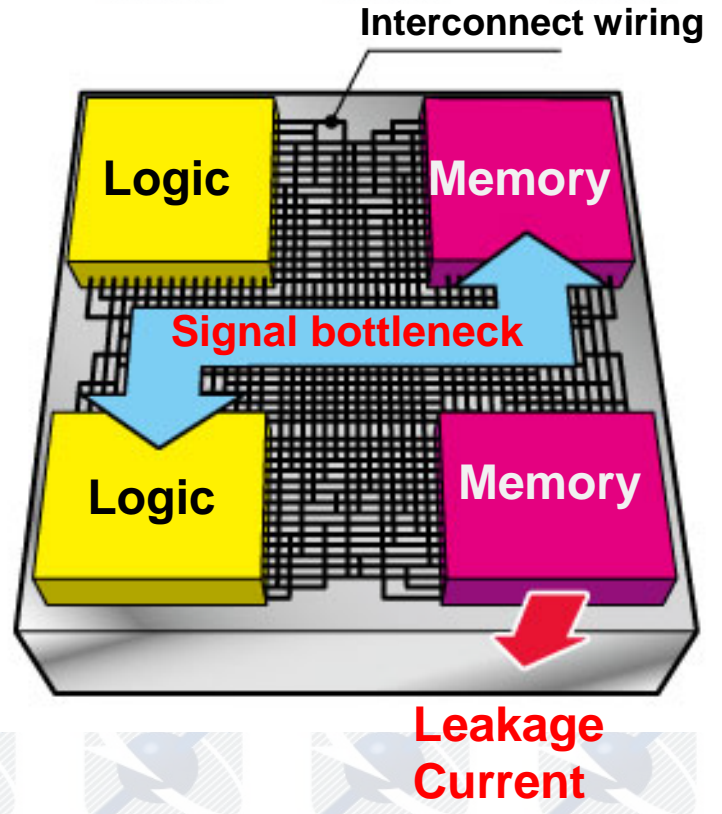
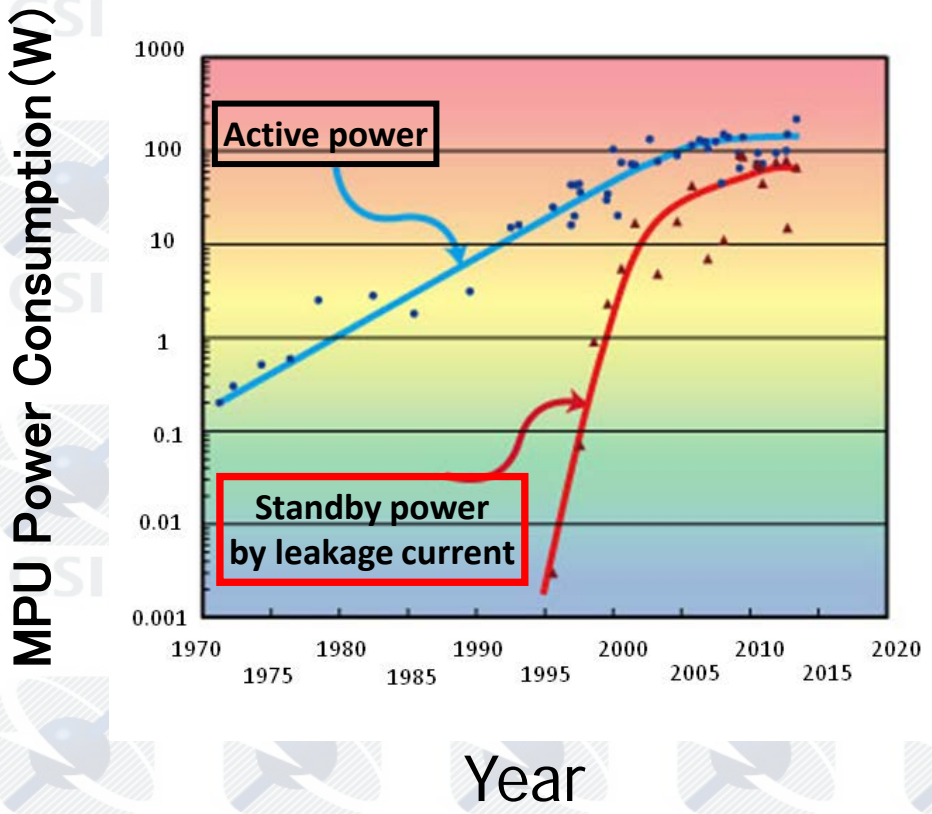
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1. Introduction
2. Magnetic Tunnel Junction (MTJ)
3. Electrical Switching of MTJ
4. Three-terminal Domain Wall Device
5. Summary

Challenges VLSI Technology Face

Power Consumption

Interconnection Delay

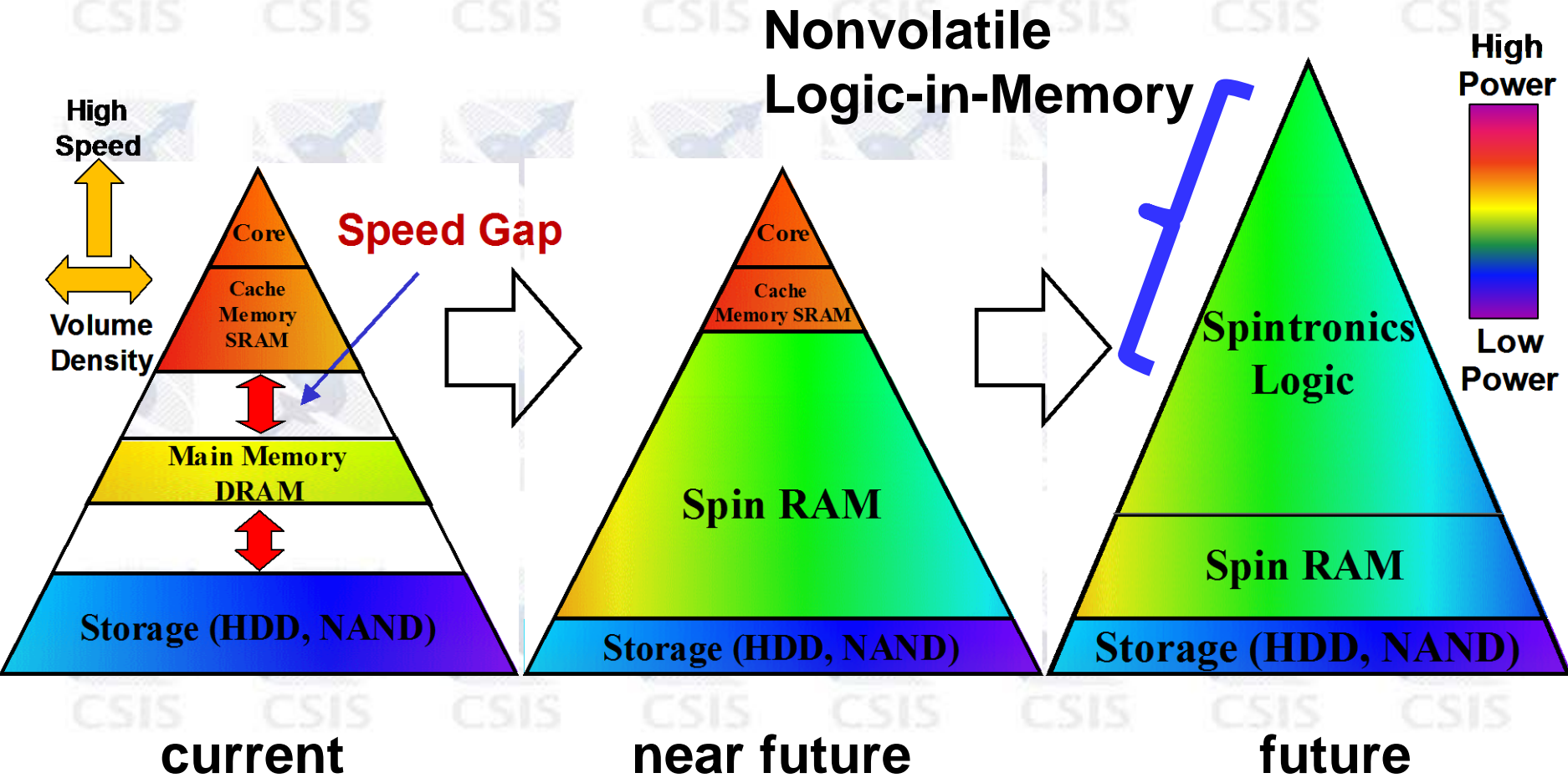




Nonvolatile working memory is in need

Features of non-volatile for memory device	Flash	FRAM	Spin Device
Access Speed	△	○	○
Non destructive Read	○	△	○
Write Endurance	×	△	○
Scalability	○	△	○
Operation Voltage	×	△	○

System (Memory) Hierarchy

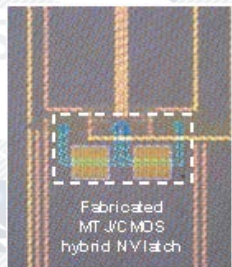


Magnetic tunnel junction based memory elements to counter **dynamic** and **static** power, and **interconnection delay**

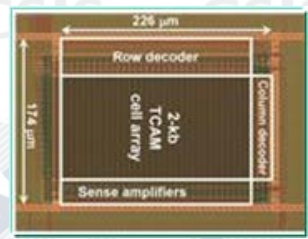
Non-volatile CMOS VLSIs with spintronics



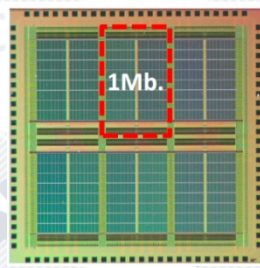
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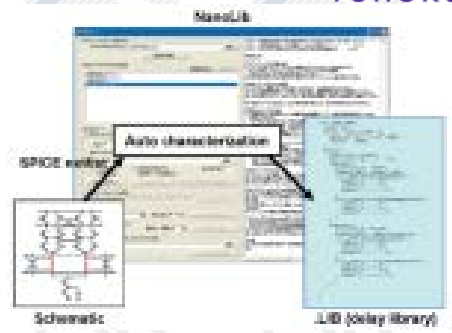
600MHz
MTJ/CMOS Latch
(Fastest nonvolatile latch)
(IEDM 2011)



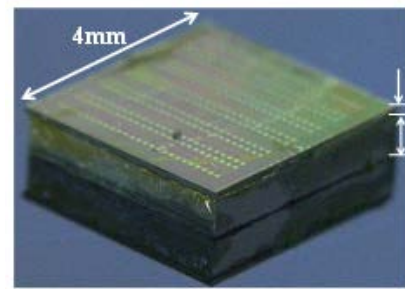
Nonvolatile TCAM
(Most compact TCAM cell, 4T-2MTJ)
(VLSI 2011)



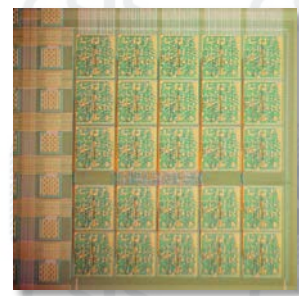
1Mb Array Three Terminal DW Cell
(High endurance)
(VLSI 2012)



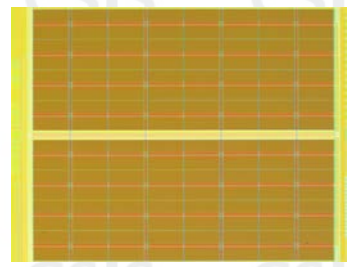
First Auto Design Tool for Spintronics CMOS
(2011)



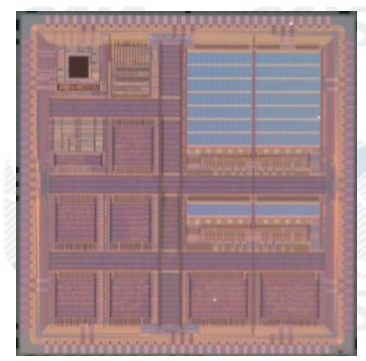
Nonvolatile FPGA with TSV
(First 3D Spintronics CMOS Processor)
(VLSI 2012)



Nonvolatile GPU
(Largest Scale Spintronics Random Logic 500kgate/chip)
(ISSCC 2013)



1.5nsec / 1Mbit
Embedded MRAM
(Fastest nonvolatile 1Mbit memory)
(VLSI 2013)

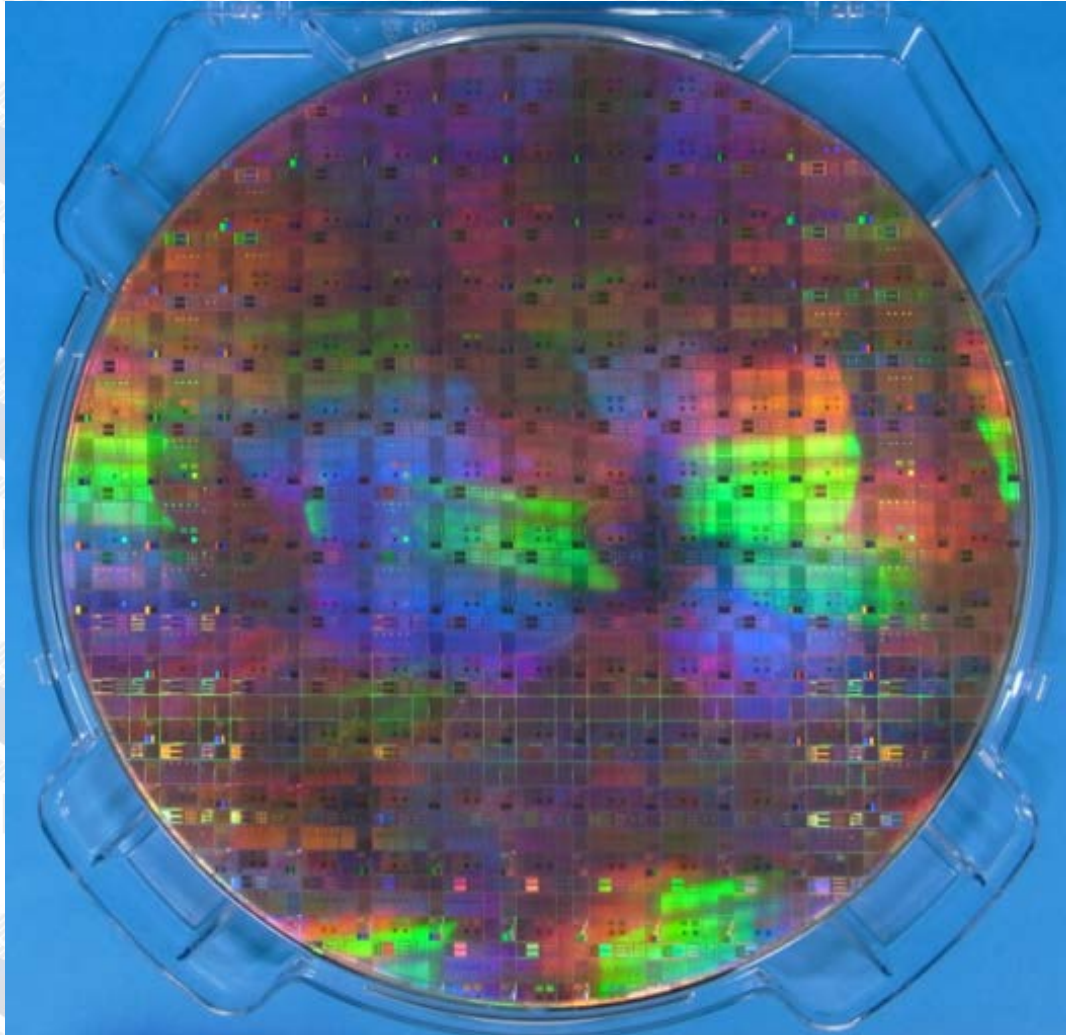


Nonvolatile microcomputer
(First nonvolatile microcomputer)
(ISSCC 2014)

On 300 mm wafers



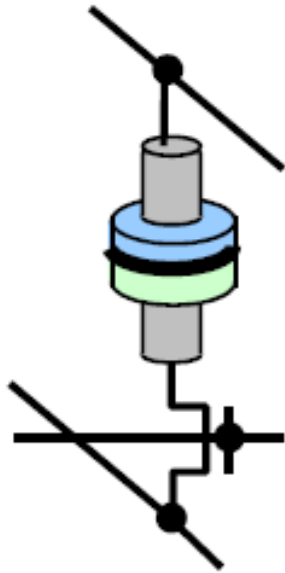
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Magnetic Tunnel Junctions (MTJs)

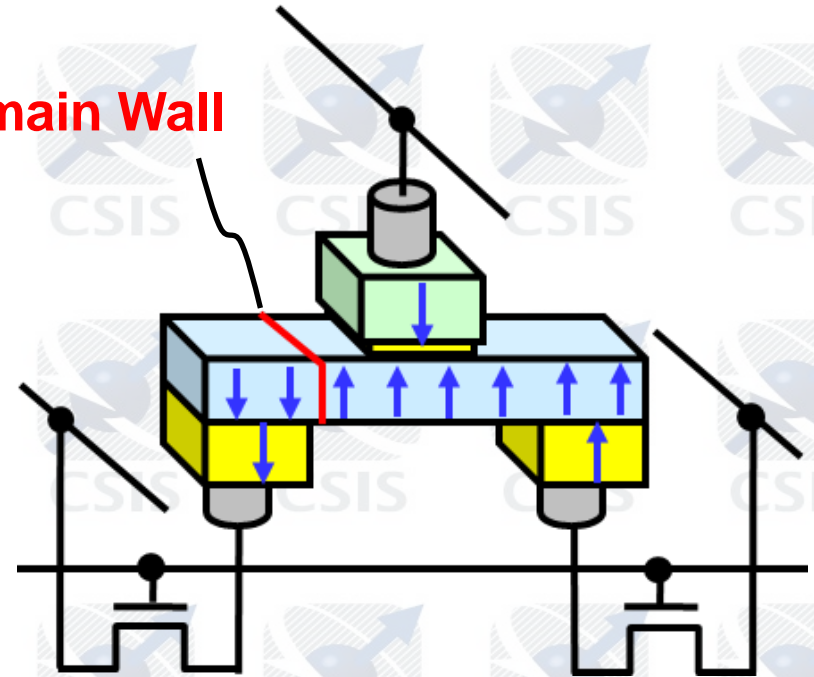


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two terminal

Domain Wall



three terminal

Nonvolatile, fast, low voltage and high endurance



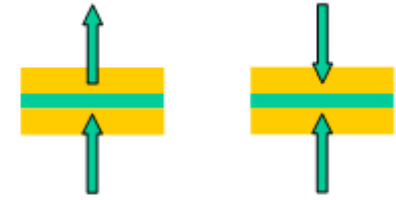
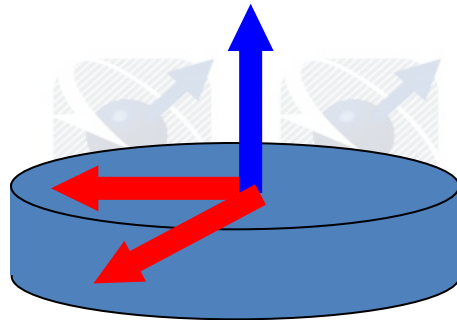
Switching Current I_{c0} and Energy Barrier $\Delta = E/k_B T$

perpendicular

$$E = \left(\frac{1}{2} M_s H_K \right) V = \underline{K_{eff}} V$$

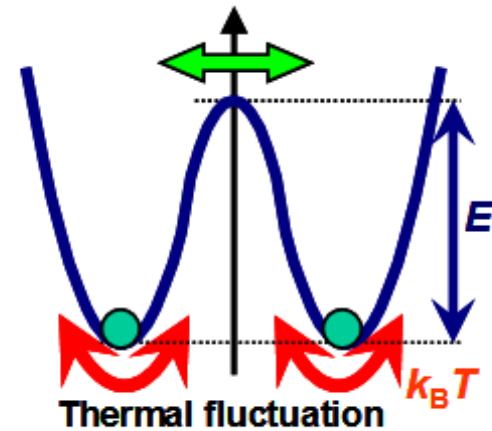
$$I_{c0} = \frac{2\alpha\gamma e}{\mu_B g} \left(\underline{K_{eff}} V \right)$$

$$\propto \alpha E$$



Parallel

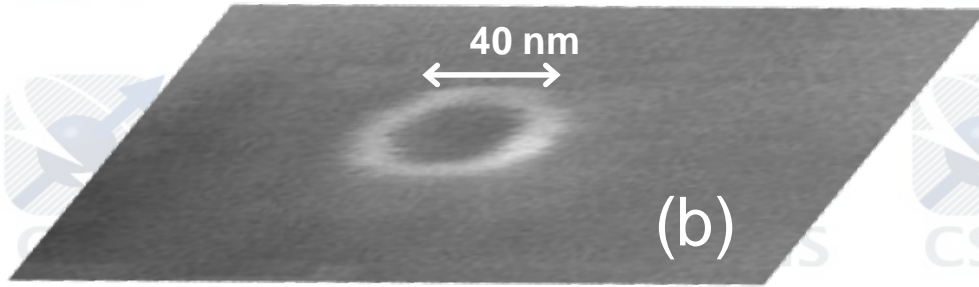
Antiparallel



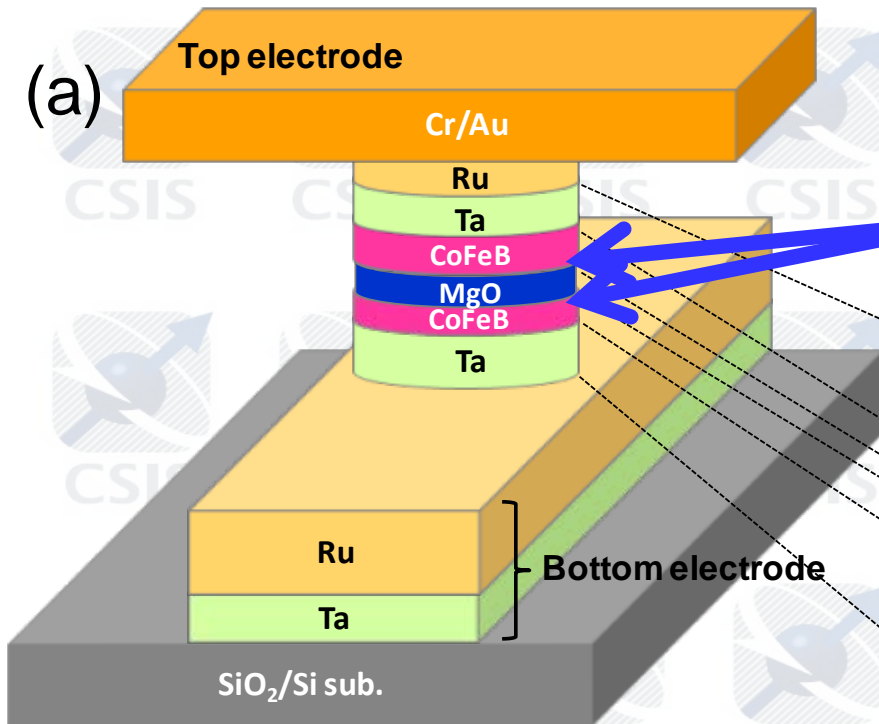
Thermal fluctuation $k_B T$



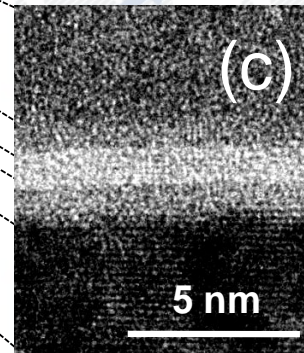
Perpendicular MgO-CoFeB MTJ



$J_{C0} = 3.8 \text{ MA/cm}^2$
($I_{C0} = 48 \mu\text{A}$)
 $E/k_B T \sim 40$
TMR ratio = 110%
 $T_a = 350^\circ\text{C}$

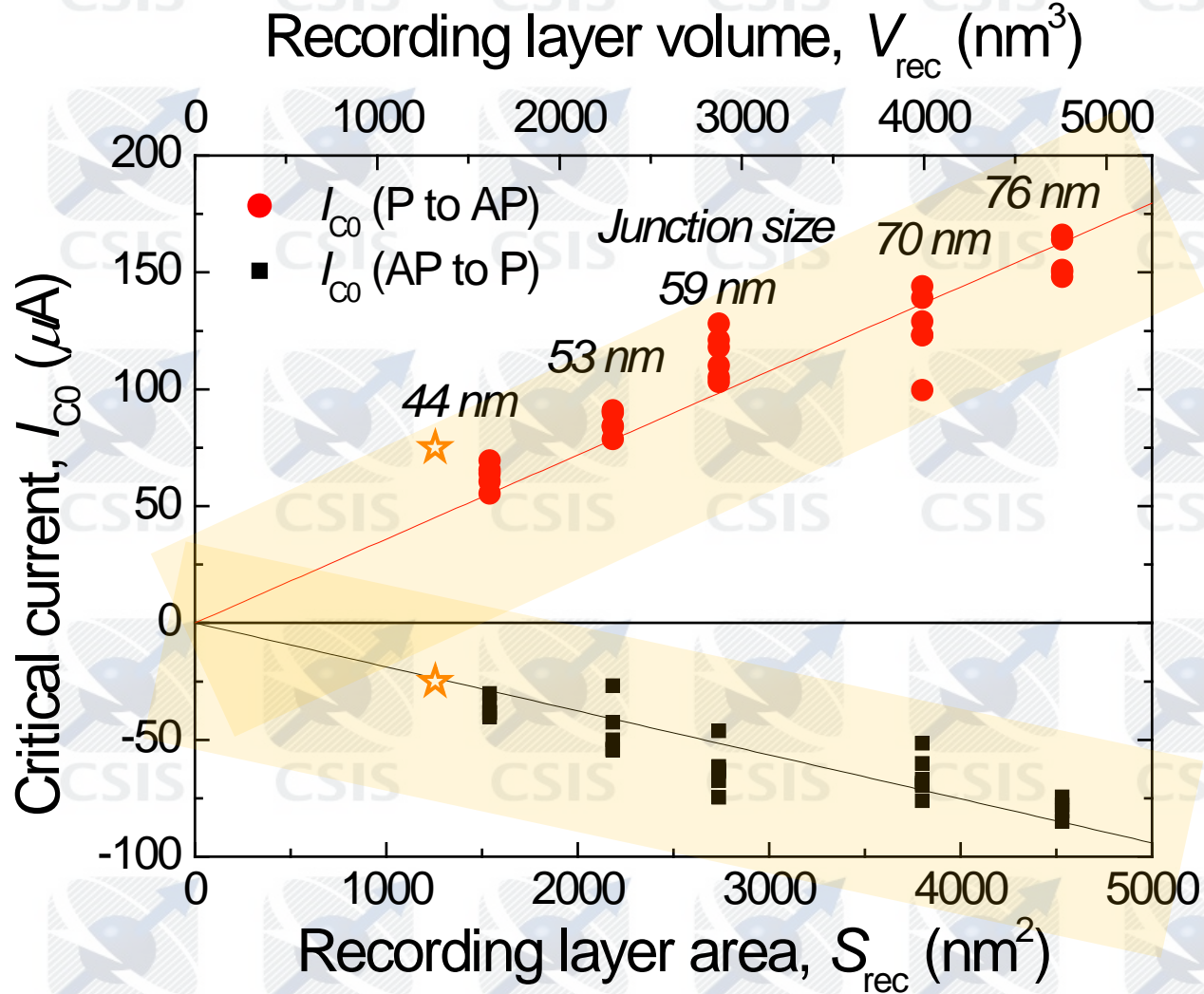


interface perpendicular anisotropy



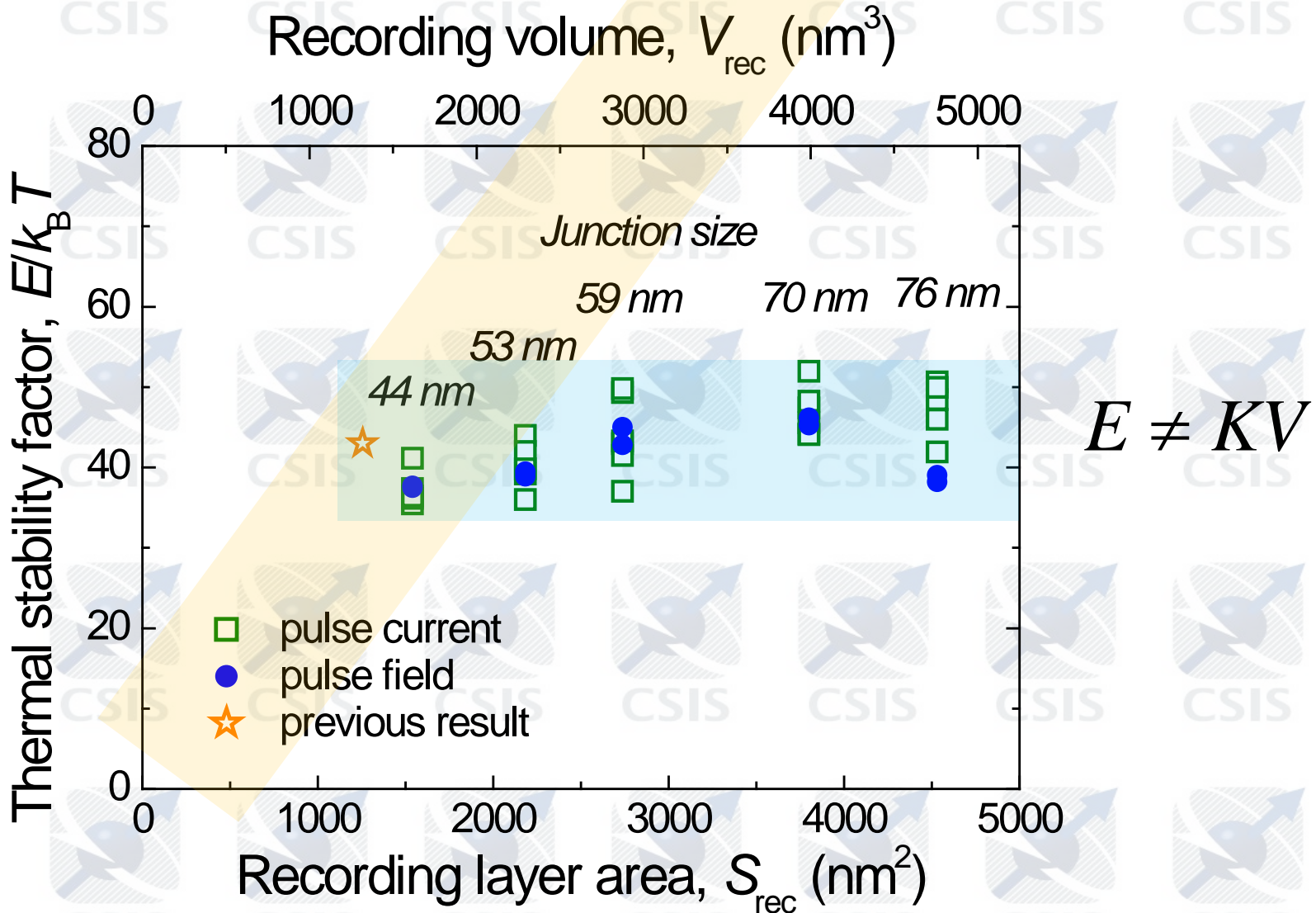


Size dependence of I_{c0}



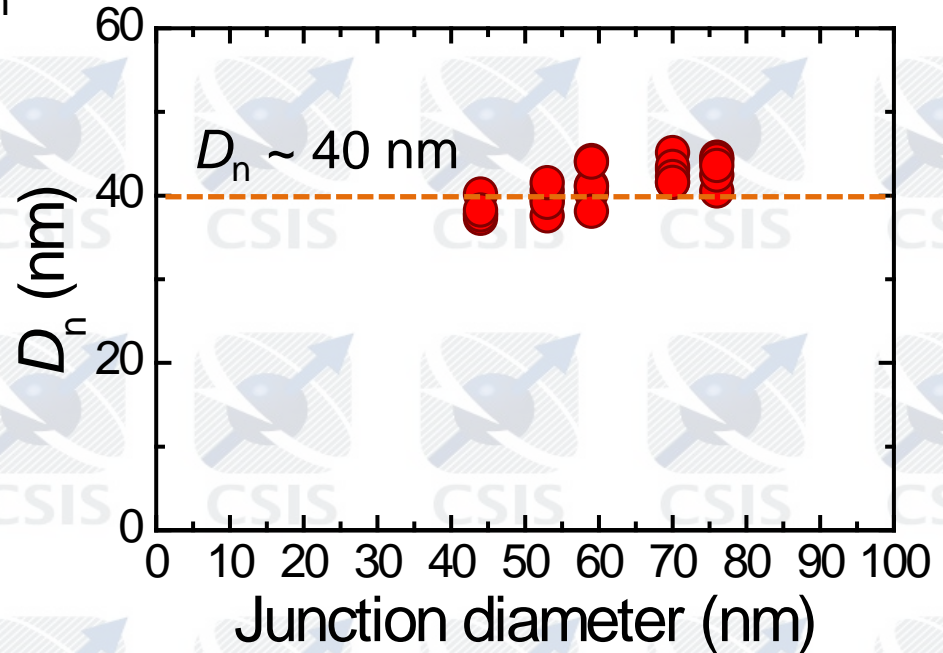
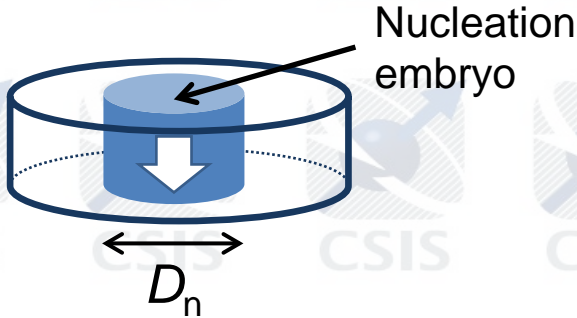


Size dependence of $\Delta = E/k_B T$





Nucleation diameter



$$\Delta = \frac{K_{eff} \left(D_n / 2 \right)^2 t}{k_B T}$$

Our results along with others suggest $D_n \approx \delta_w = \pi \sqrt{\frac{A_s}{K_{eff}}}$



E and layer thickness

$$E = K_{eff} V = K_{eff} \pi \left(D_n / 2 \right)^2 t$$



$$E \approx K_{eff} \pi \left(\delta_w / 2 \right)^2 t$$

$$= \frac{\pi^3 A_s t}{4}$$

$$\propto t$$

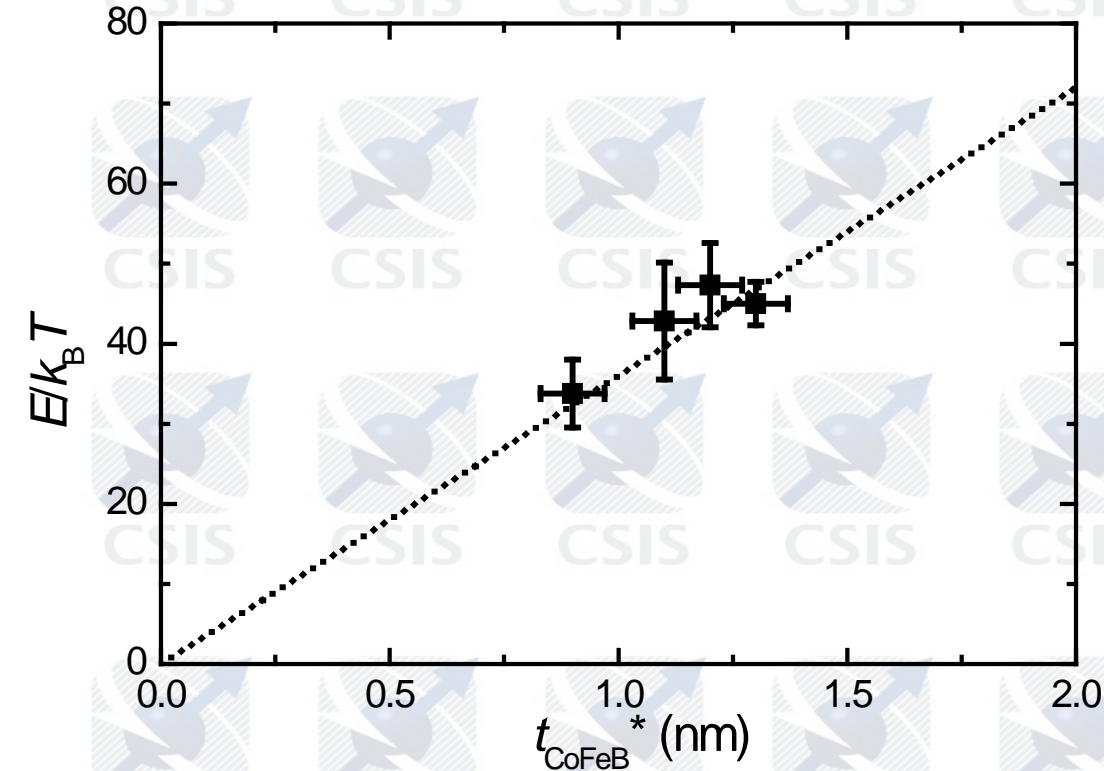
H. Sato *et al.*, IEEE Magn. Lett. **3**, 3000204 (2012).

$$\left(K_{eff} = \frac{K_i}{t} + K_b - \frac{M_s^2}{2\mu_0} \right)$$

CoFeB thickness dependence of $E/k_B T$



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- t_{CoFeB}^* ($= t_{\text{rec}}$) was determined by subtracting magnetically dead layer (0.4 nm) at CoFeB/Ta interface.

$$E \approx K_{\text{eff}} \pi (\delta_w / 2)^2 t_{\text{rec}}$$
$$= \pi^3 A_s t_{\text{rec}} / 4$$

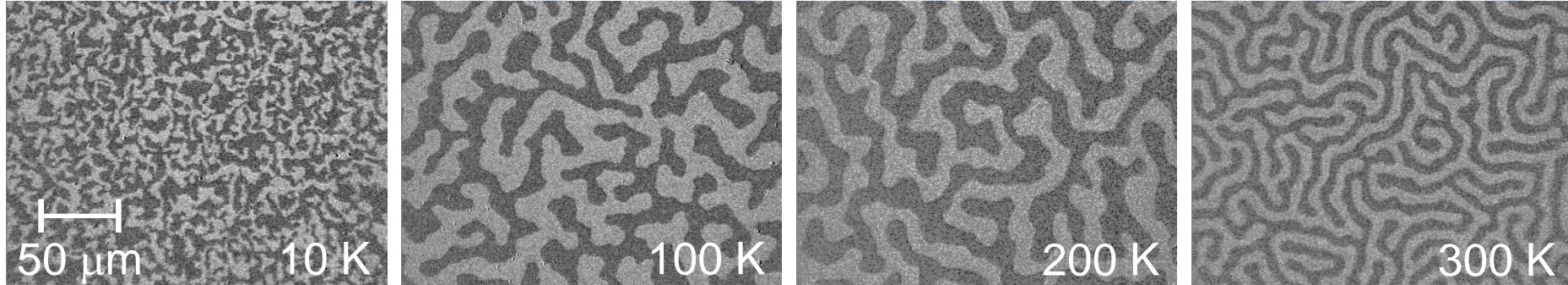
- A linear relationship between $E/k_B T$ and t_{rec}
- The slope of $36 \text{ nm}^{-1} \Rightarrow A_s^* \approx 19 \text{ pJ/m}$



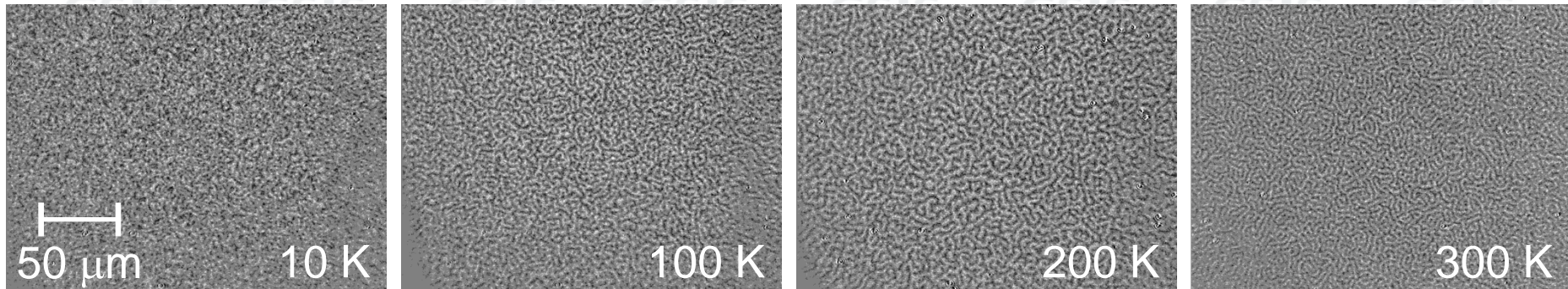
Domain patterns of CoFeB

Magneto-optical Kerr effect (MOKE) images after demagnetization

sample A ($t = 1.1$ nm), as-deposited



sample B ($t = 1.3$ nm), annealed at 350°C



$T \geq 100$ K

- Domain walls moved smoothly
- Labyrinth patterns were formed

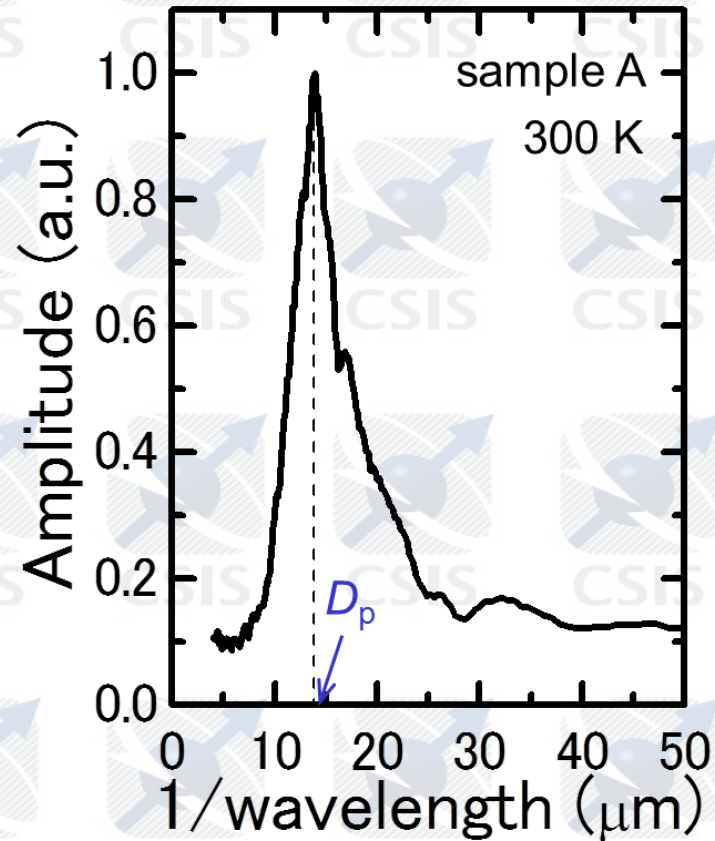
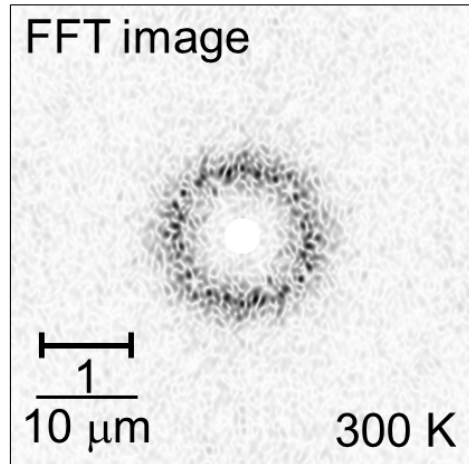
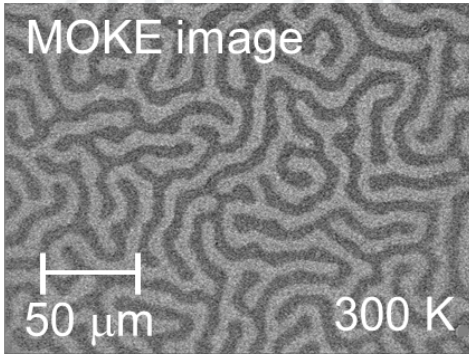
$T \leq 50$ K

- Domain walls were strongly pinned
- Complex patterns were formed



Domain patterns of CoFeB

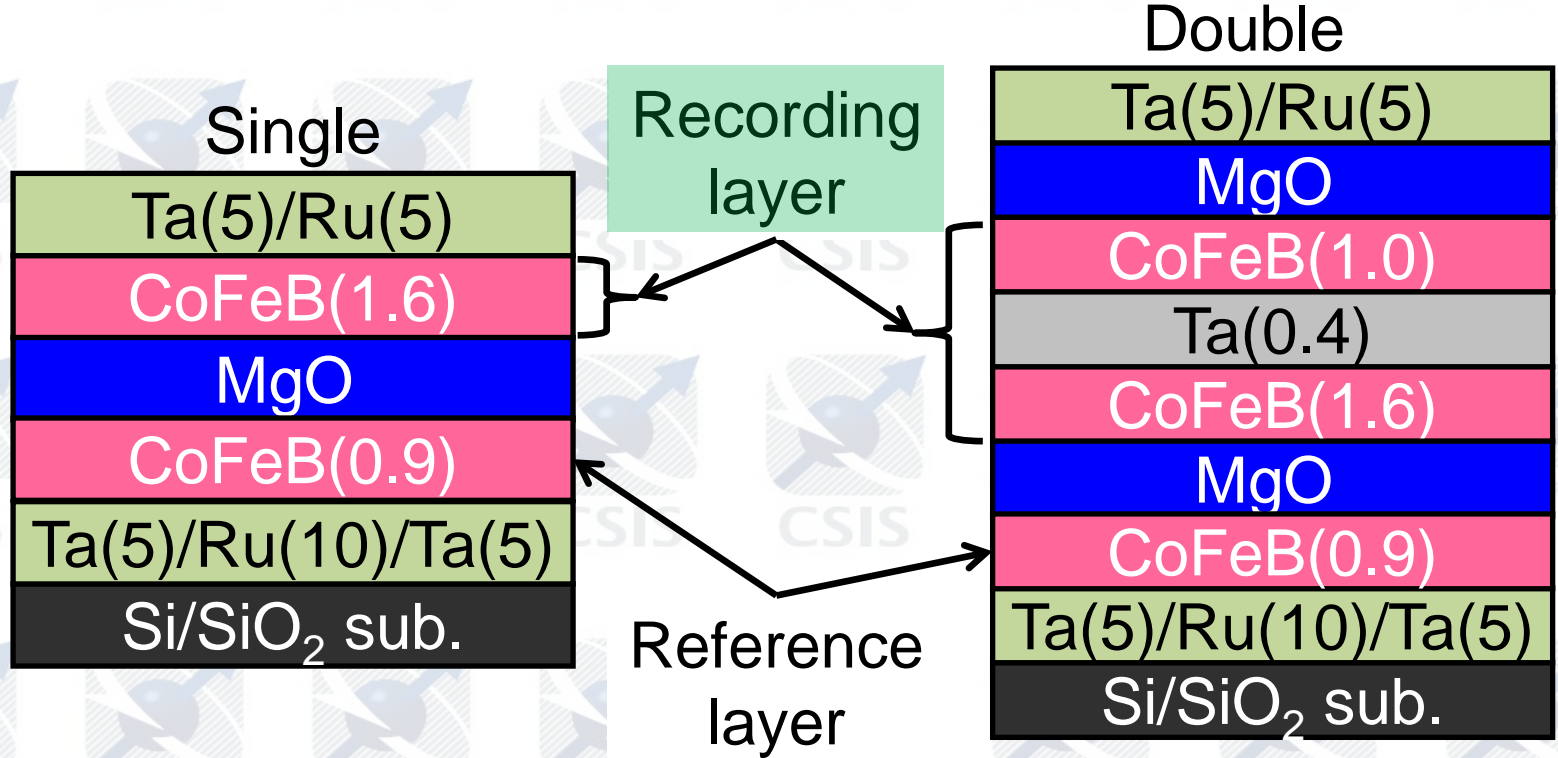
sample A



	A_s (pJ/m) at 300 K	δ_w (nm) at 300 K
sample A ($t = 1.1$ nm, As-deposited)	8.4	43
sample B ($t = 1.3$ nm, 350°C)	31	67



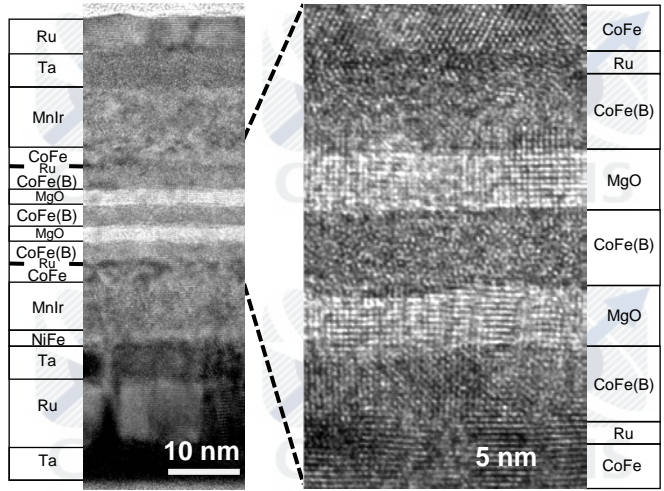
Double interface structure



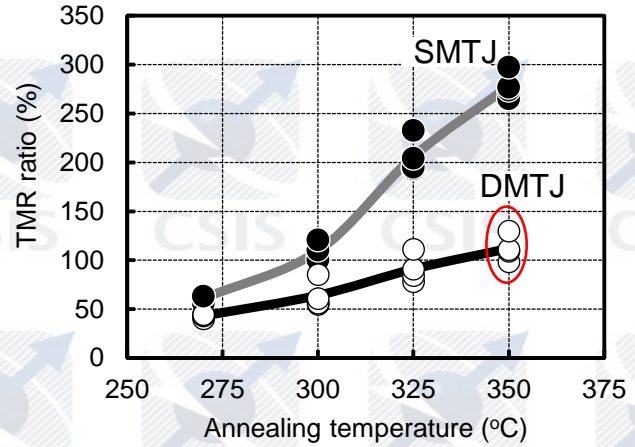
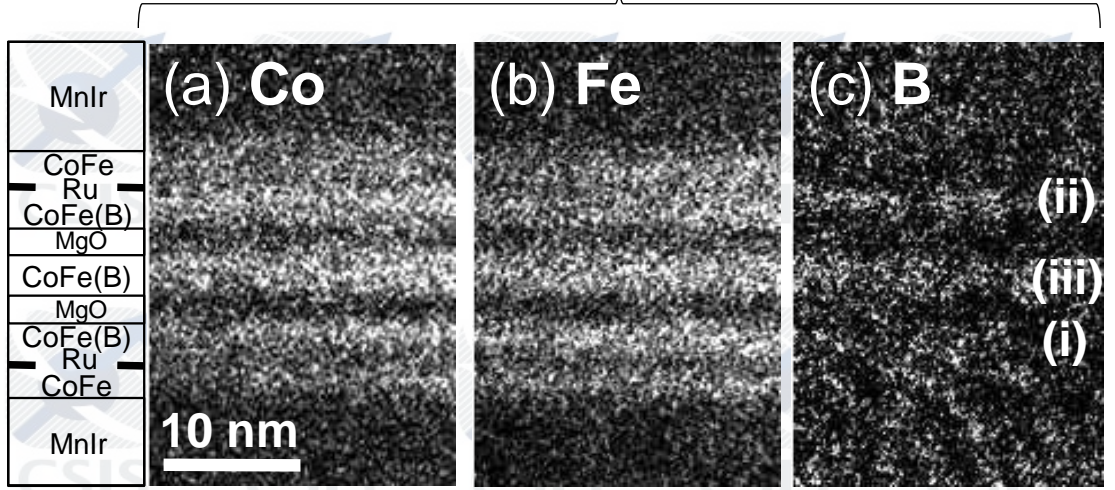
(70 nm ϕ)	Single	Double
$E/k_B T$	51	95
J_{C0} (MA/cm ²)	3.3	3.2

Double MgO

Double-MgO MTJ (DMTJ)
annealed at $T_a = 350\text{ }^\circ\text{C}$



EELS element map





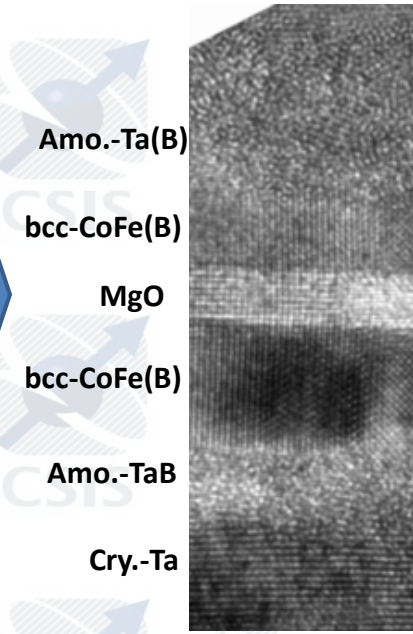
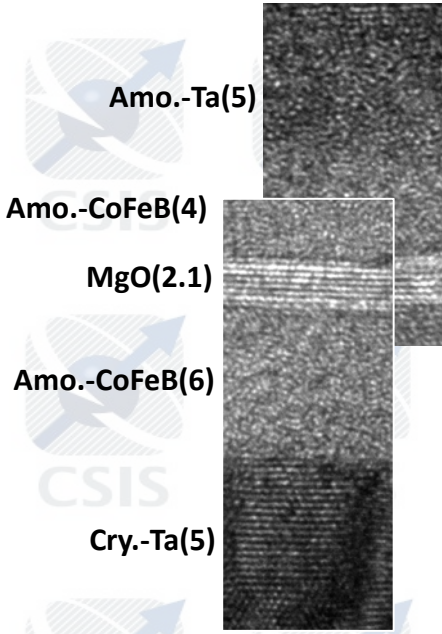
Single MgO

As-depo.

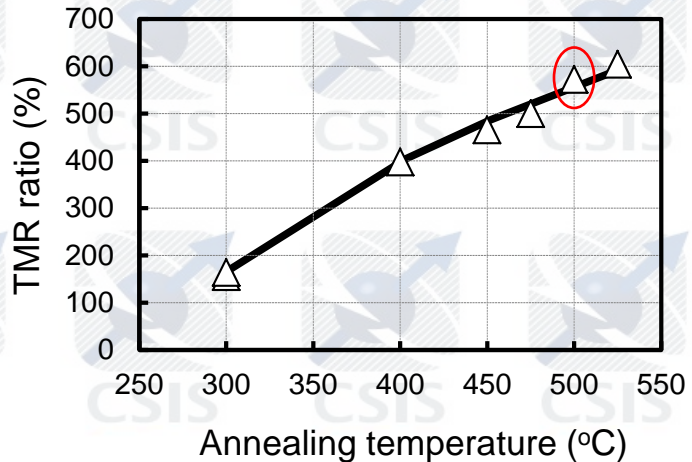
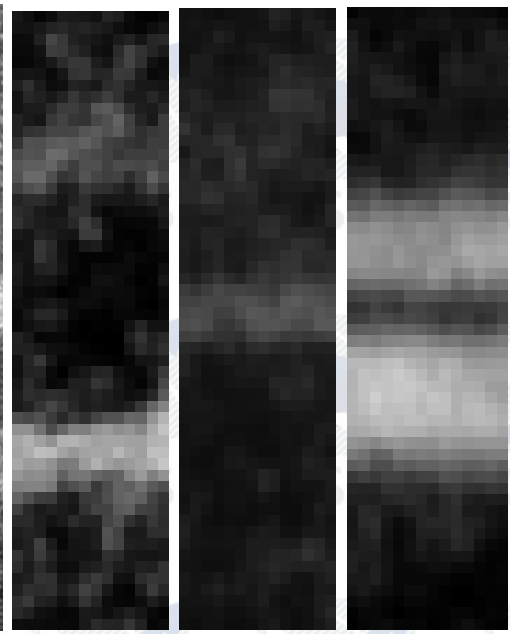
$T_a = 500\text{ }^\circ\text{C}$

EELS element map

B O Fe



5 nm



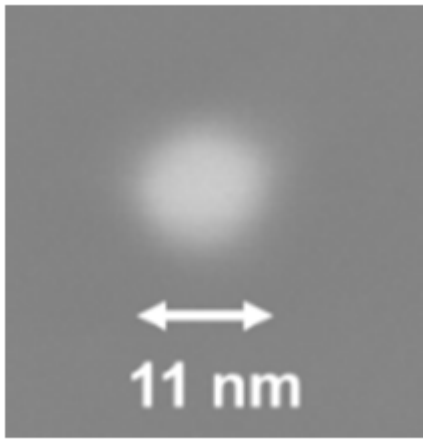
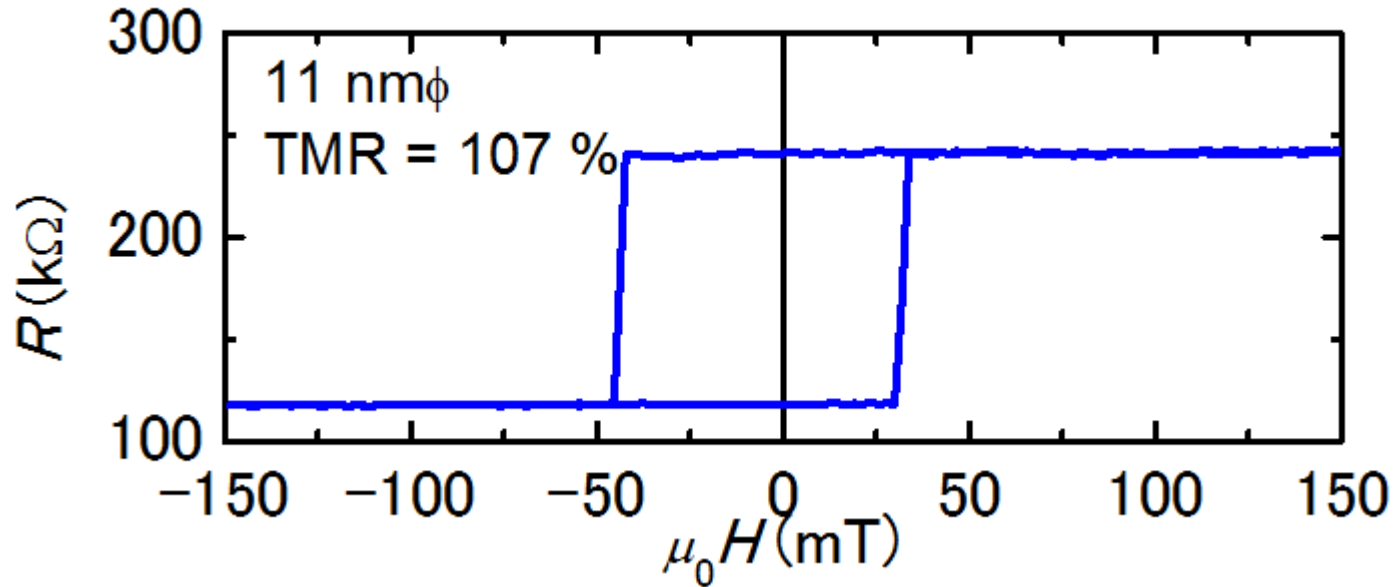
S. Ikeda et al., Appl. Phys. Lett. 93(2008) 082508.

S. V. Karthik et al., J. Appl. Phys. 106 (2009) 023920.

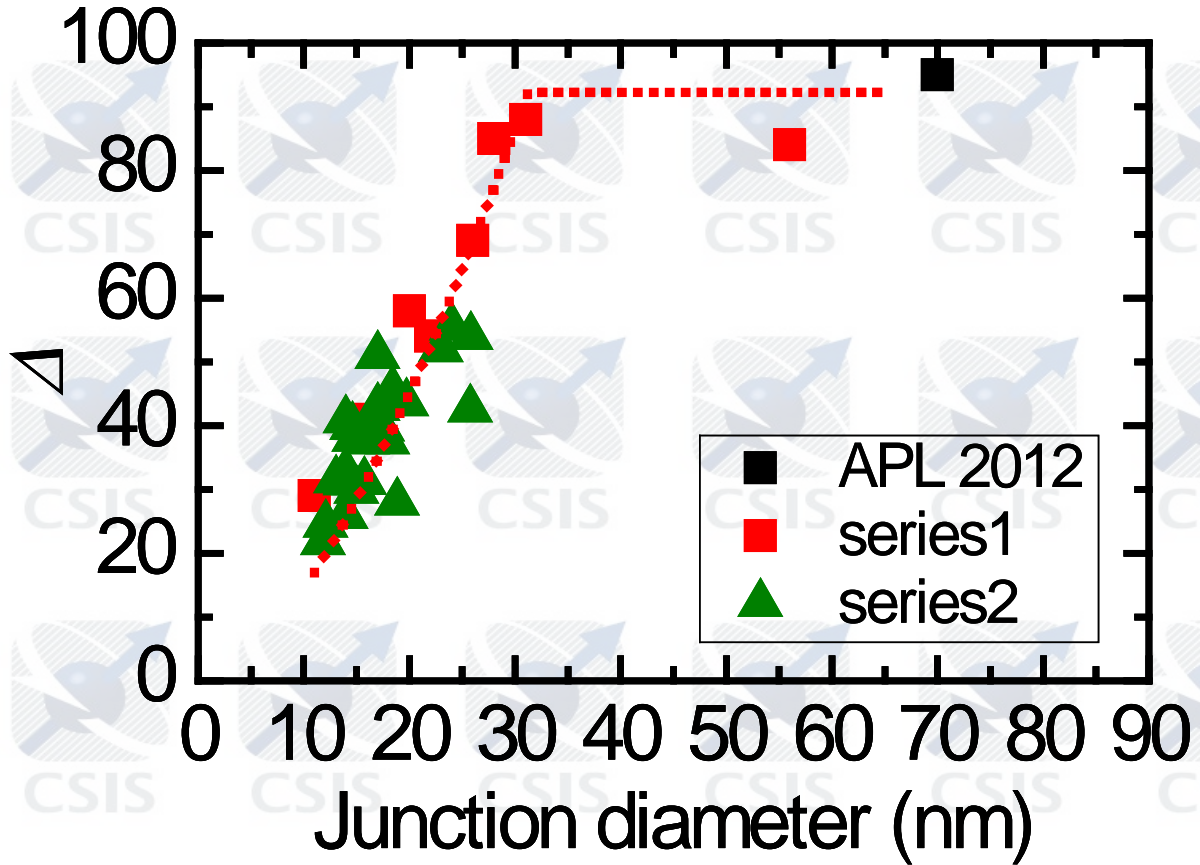
11 nm ϕ MTJ (smallest MTJ to date)



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Device size dependence of Δ



Dotted line reproduces the trend well
Interface engineering to further enhance Δ

H. Sato *et al.*, *Appl. Phys. Lett.* **101**, 022414 (2012).
H. Sato *et al.*, IEDM 2013, p. 3.2.1.
H. Sato *et al.* *Appl. Phys. Lett.* **105**, 062403 (2014)

Demagnetization coefficient N



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$$E = K_{eff} t \left(\frac{D}{2} \right)^2 \pi$$

$$K_{eff} = \frac{K_i}{t} + K_b - \frac{M_s^2}{2\mu_0} \underline{(N_z - N_x)}$$



Reported properties of nano p-MTJs

Material	Size (nm)	TMR ratio (%)	I_{C0} or I_C (μA)	$E/k_B T$	Ref
CoFeB	40	124	49	43	[1]
CoFeB	17x40	100 (CIPT)	50	35	[2]
CoFeB	20	57	29	29	[3]
undisclosed	30	73	25	61	[4]
CoFeB	27	130	12	80	[5]
			19	147	
undisclosed	15	-	~ 0.6 V	~ 42	[6]
CoFeB/Ta/ CoFeB	20	127	24	58	[7]
	15	101	22	41	[7]
	11	107	13	28	[7]

[1] S. Ikeda *et al.*, Nature Mater. 9, 721 (2010).

[2] W. Kim *et al.*, 2011 IEDM, p24.1.1

[3] M. Gajek *et al.*, Appl. Phys. Lett. 100, 132408 (2012).

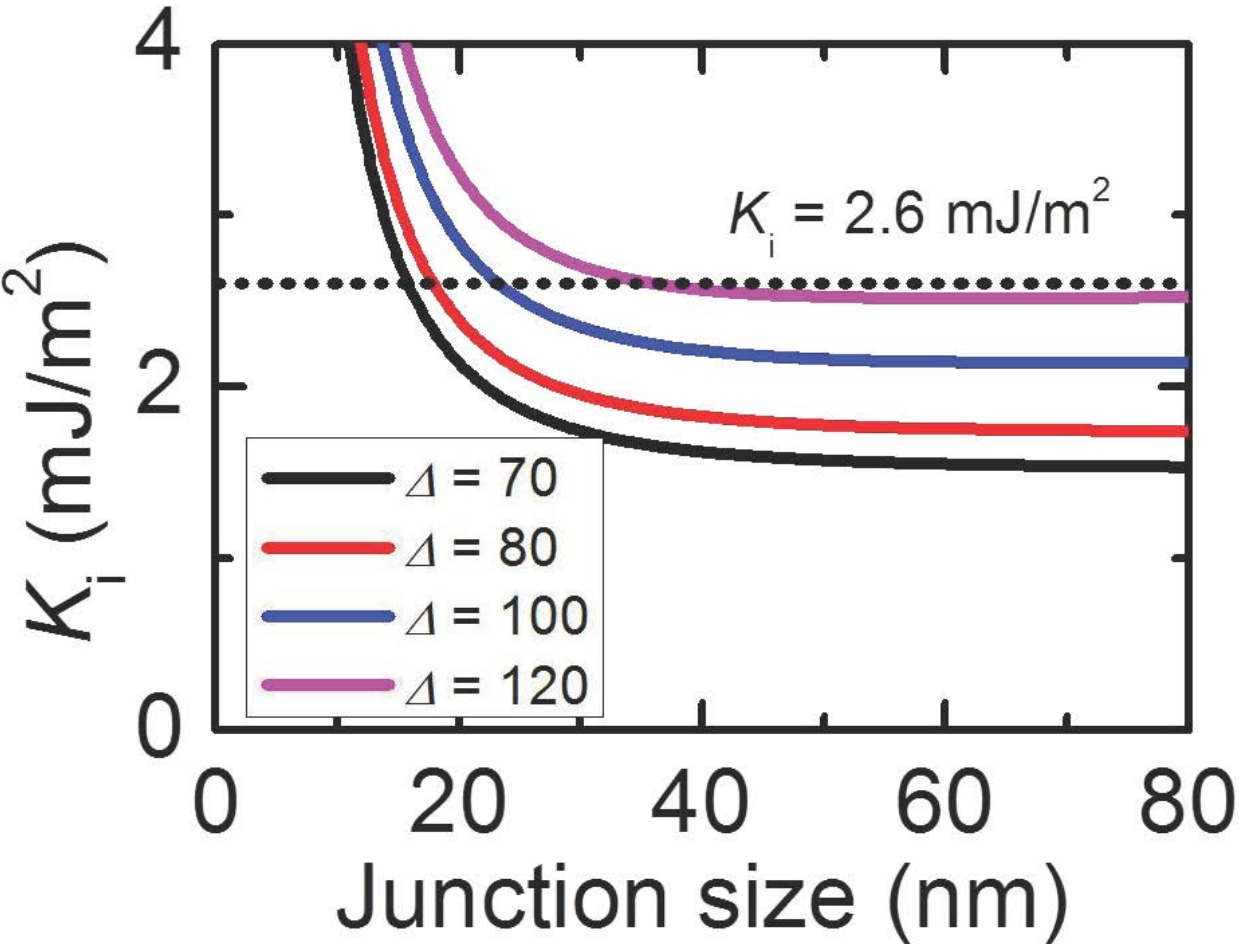
[4] E. Kitagawa *et al.*, 2012 IEDM, p. 29.4.2.

[5] L. Thomas *et al.*, J. Appl. Phys. 115, 172615 (2014).

[6] J. H. Kim *et al.*, 2014 VLSI Tech., P.76.

[7] H. Sato *et al.*, 2013 IEDM, p. 61., Appl. Phys. Lett. 2014

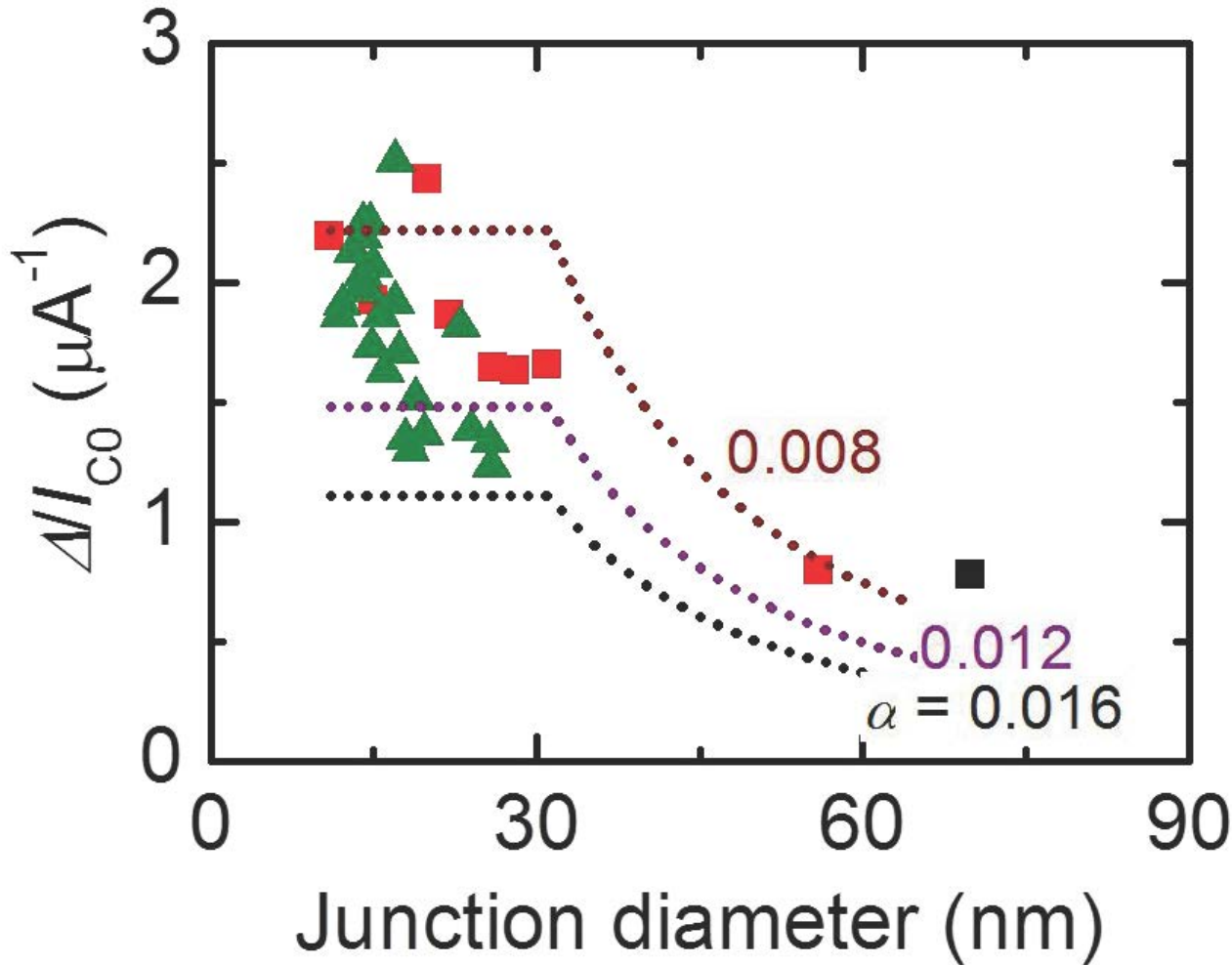
Interface anisotropy – junction size, K_i and Δ



Size dependence of ΔI_{C0}



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H. Sato *et al.*, *Appl. Phys. Lett.* **101**, 022414 (2012).

H. Sato *et al.*, *IEDM 2013*, p. 3.2.1.

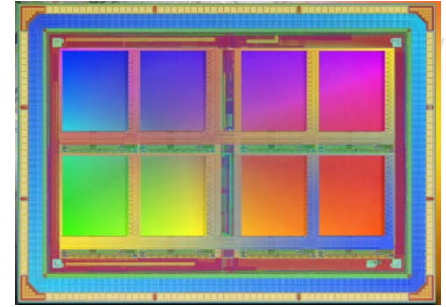
H. Sato *et al.*, *Appl. Phys. Lett.* **105**, 062403 (2014)



Magnetization manipulation by

Magnetic field

write/read heads for HDD
1st generation MRAM



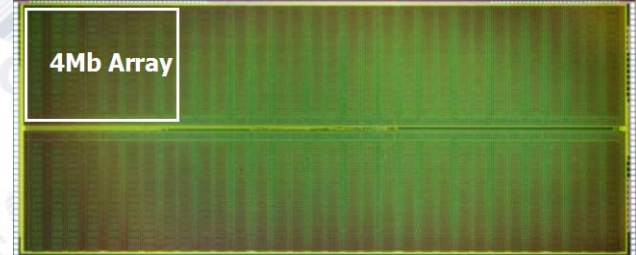
Spin current

L. Berger, J. Appl. Phys. 55, 1954 (1984).
J. Slonczewski, J. Magn. Magn. Mat. 159, L1 (1996).
L. Berger, Phys. Rev. B 54, 9353 (1996).

<http://www.hitachigst.com/>

<http://www.everspin.com/>

Spin torque MRAM
Spin torque oscillator
Race-track memory



R. Takemura *et al.*, VLSI Circ. Dig. p.84 (2009)

Electric field

Switching Energy



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Spin-transfer switching

$$\begin{aligned} VIt &= 0.25 \text{ (V)} \times 30 \text{ (\mu A)} \times 1 \text{ (ns)} \\ &= 8 \text{ (fJ)} \end{aligned}$$

Electric field switching

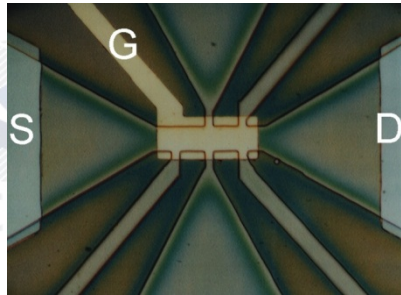
$$\begin{aligned} CV^2 &= S \times d \times \varepsilon \times E^2 \\ &= \pi \left(\frac{30 \text{ (nm)}}{2} \right)^2 \times 5 \text{ (nm)} \times 9.8 \varepsilon_0 \times (5 \text{ (MV/cm)})^2 \\ &= 0.08 \text{ (fJ)} \end{aligned}$$

Electric-field control of magnets

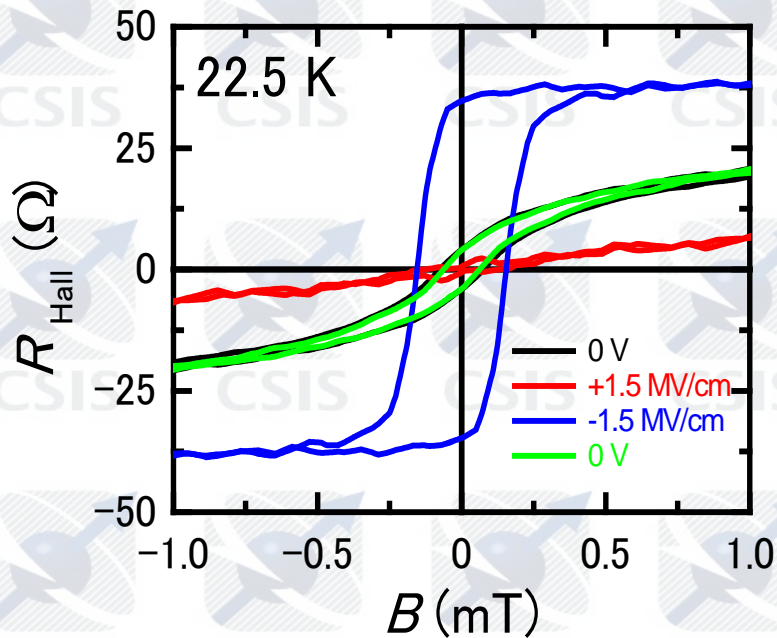


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Ferromagnetic transition

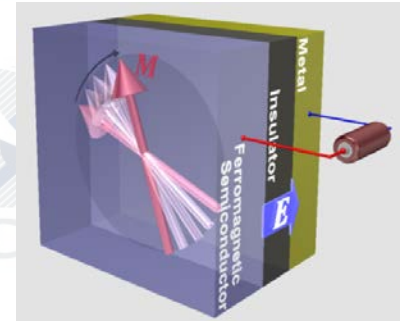


Ferromagnetic semiconductor (In,Mn)As

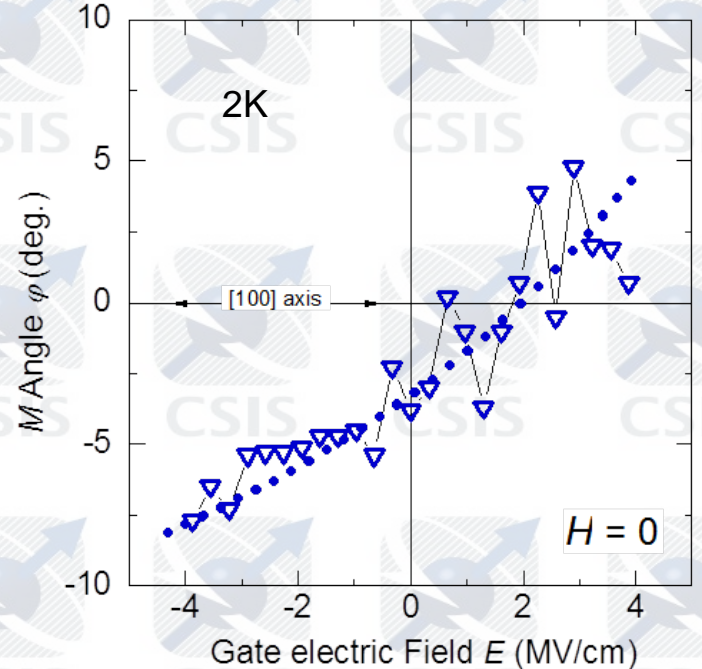


H. Ohno *et al.*, *Nature* 408, 944 (2000)

Magnetization direction



Ferromagnetic Semiconductor (Ga,Mn)As

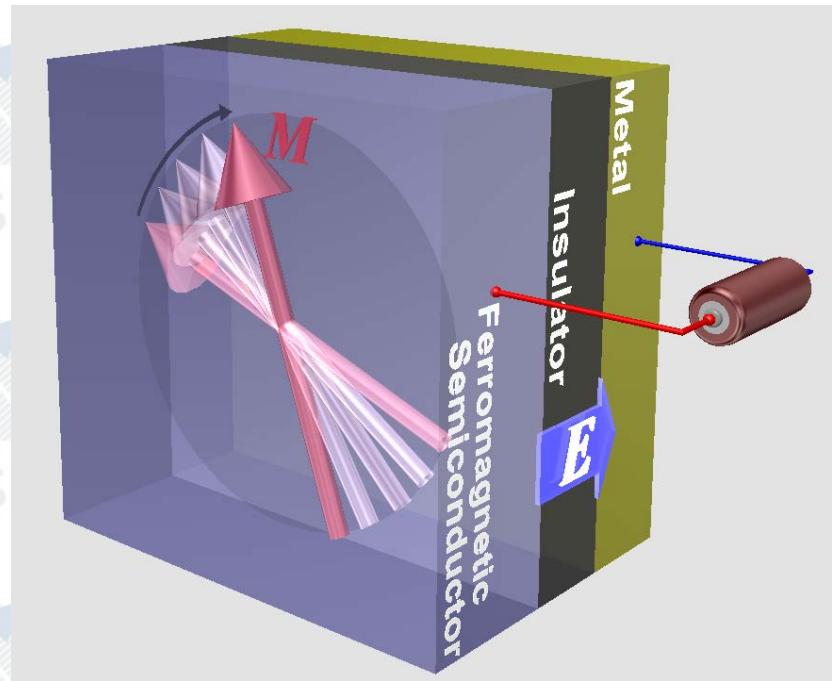


D. Chiba *et al.*, *Nature* 455, 515 (2008)

Magnetization switching by anisotropy



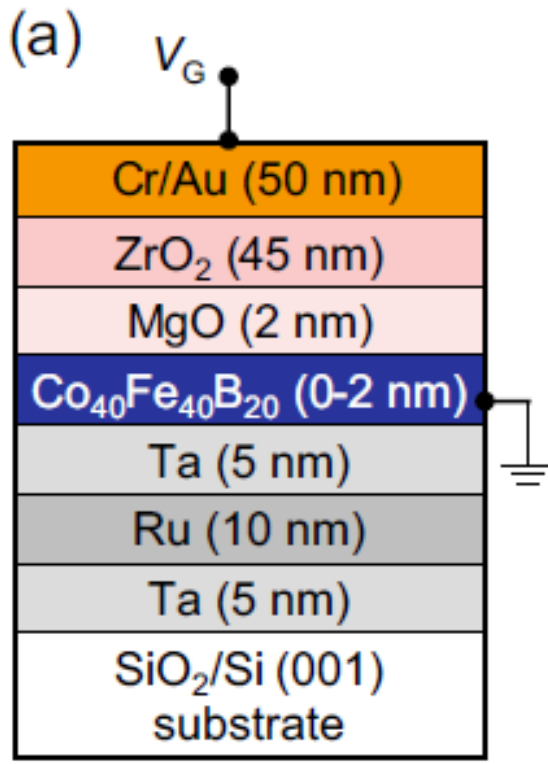
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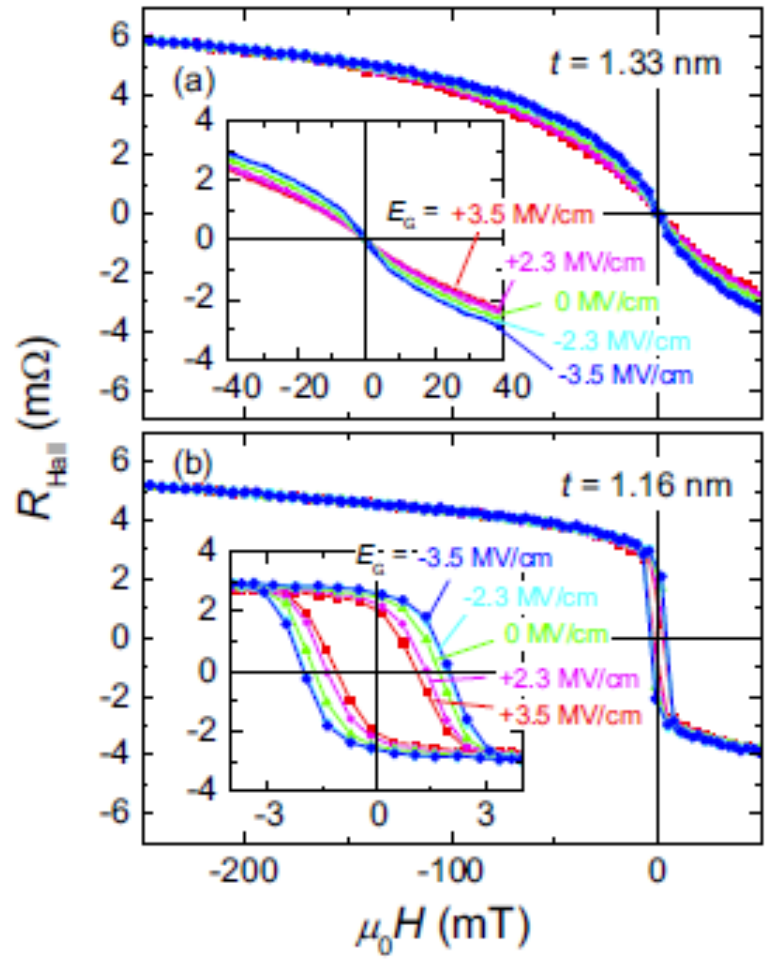
- (1) *perpendicular* to *in-plane* and back
- (2) controlling *in-plane* anisotropies

Electric-field effects on metals

Room temperature



30 $\mu\text{eV}/\text{m}^2$ per V/nm

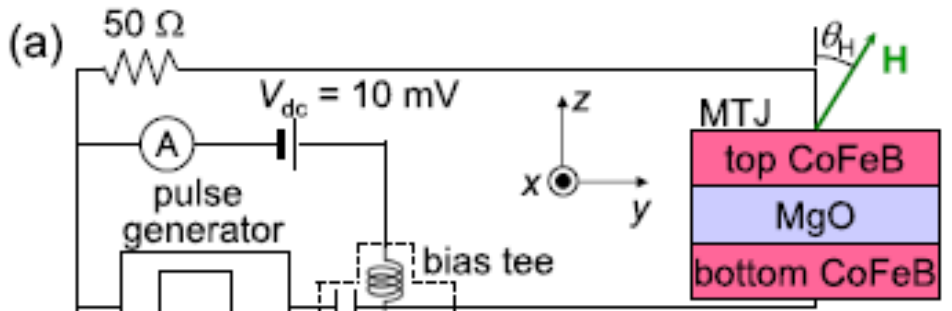
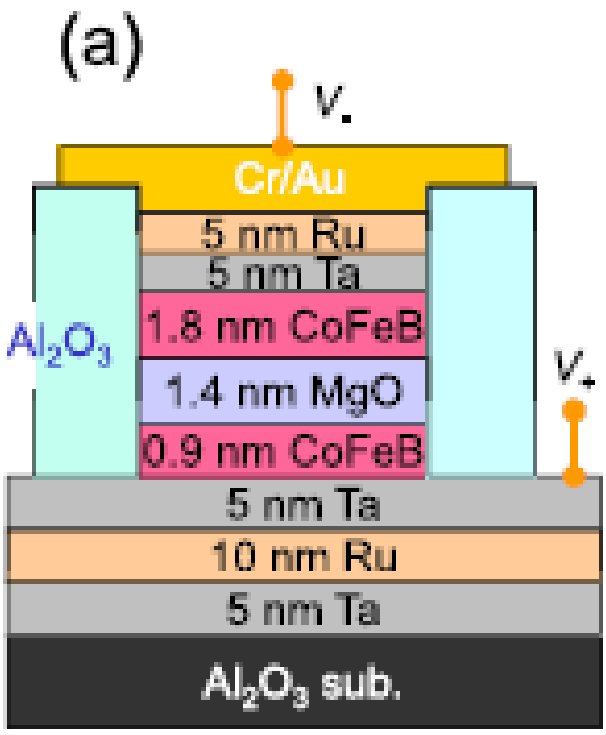


Electric-field modulation of anisotropy in CoFeB

M. Endo, S. Kanai, S. Ikeda, F. Matsukura, and H. Ohno, *Appl. Phys. Lett.* 96, 212503 (2010).

See also; FePt, FePd: M. Weisheit *et al.*, *Science* (2007). Fe/Au: T. Maruyama *et al.*, *Nature Nanotechnology* (2009).

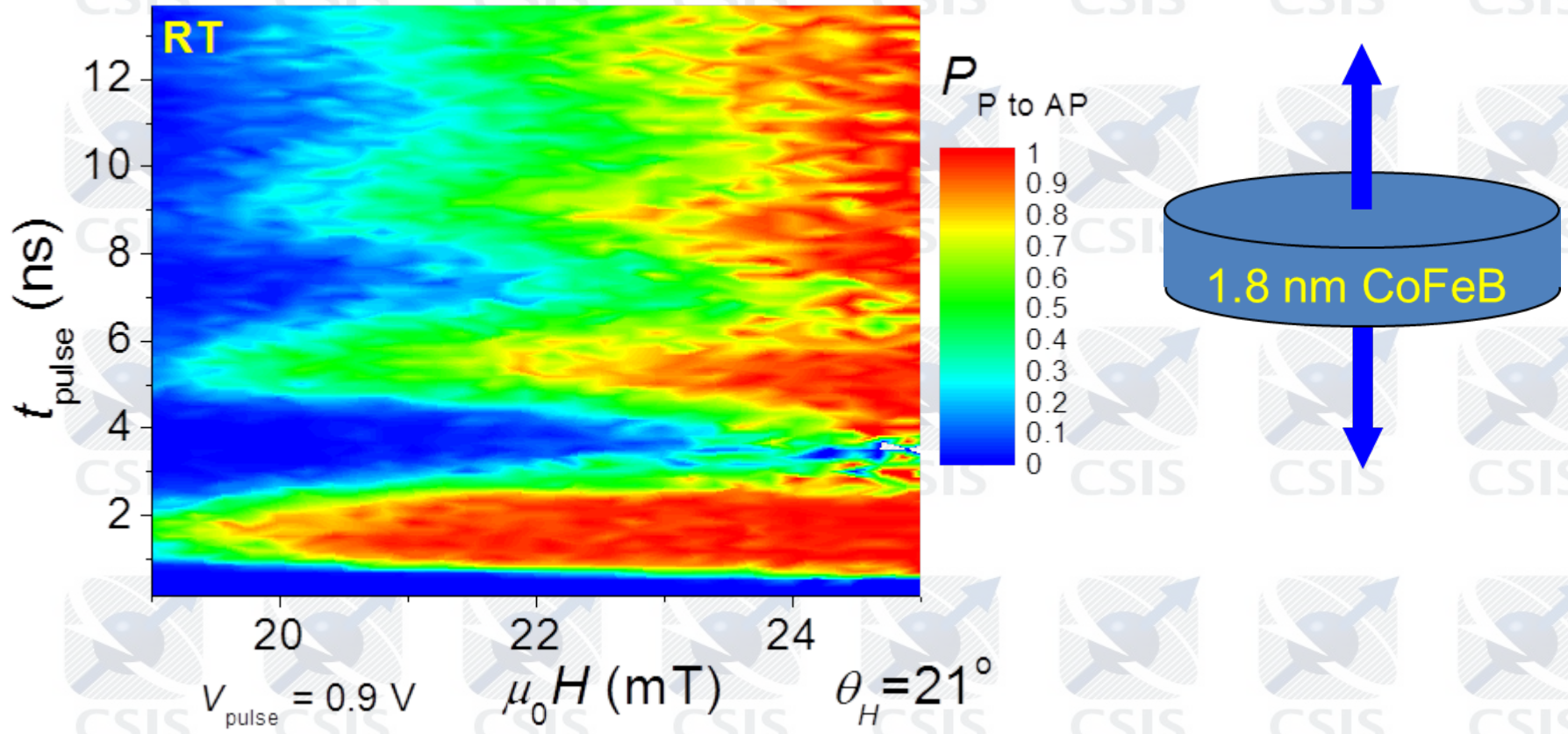
Electrical switching of *perpendicular* CoFeB



70 nm ϕ



Electrical switching of *perpendicular* CoFeB

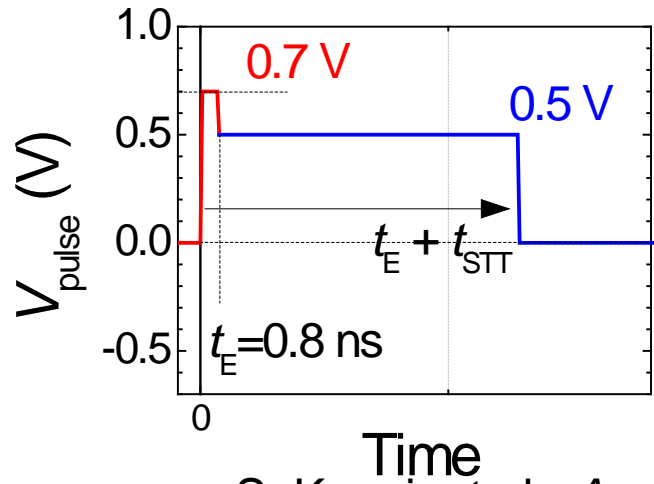
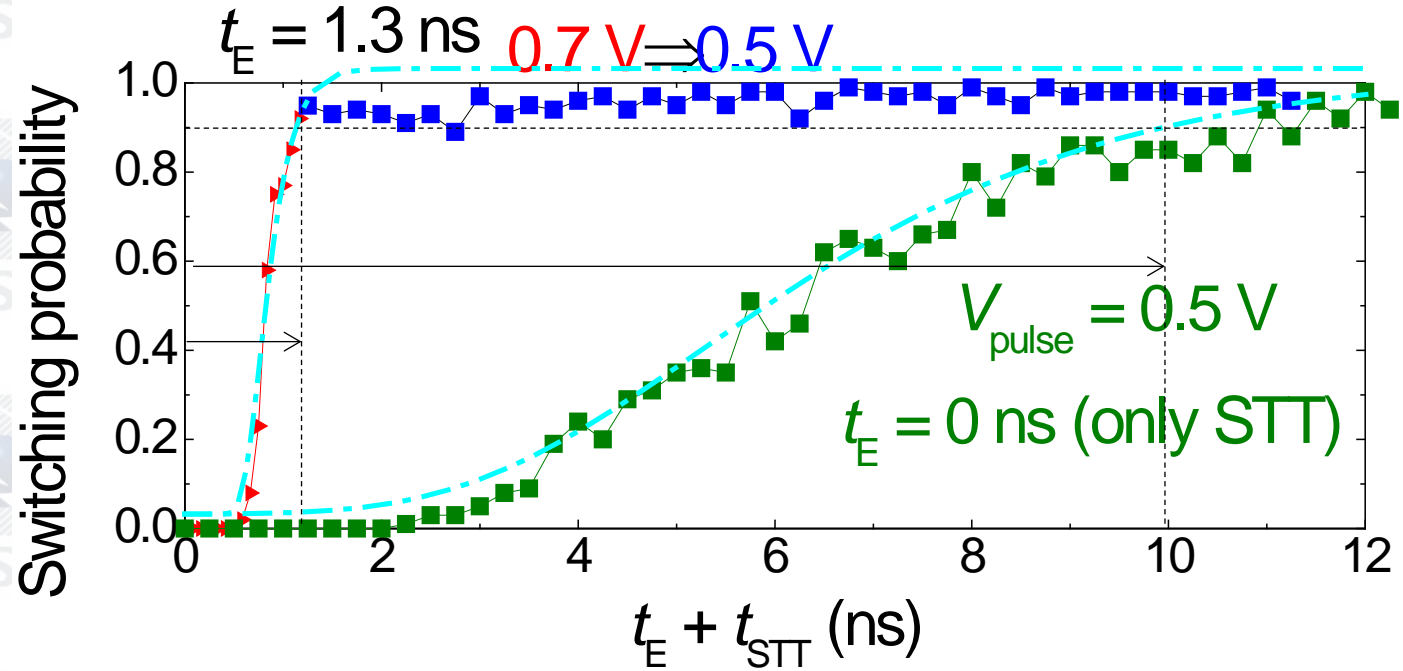


under constant magnetic field

S. Kanai, *et al.*, *Appl. Phys. Lett.* **101**, 122403 (2012)
S. Kanai, *et al.*, *Appl. Phys. Lett.* **103**, 072408 (2013)

See also Y. Shiota *et al.* *Nature Materials*, 2011 for ultrathin FeCo
W. G. Wang *et al.* *Nature Materials* 2012 for electric-field assisted switching

Electrical switching plus STT



Remaining challenges



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✓ Toggle: read before write

✓ Scaling: $E=KV$, $\Delta K_i=E/S$

✓ Configuration: two terminal vs. three

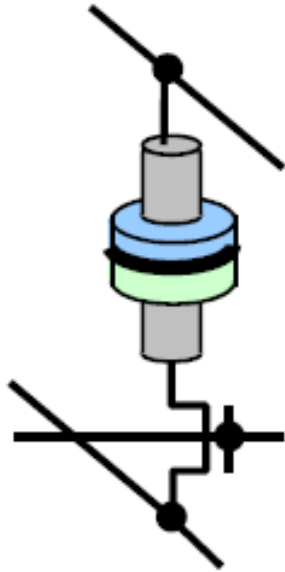
✓ Switching probability

✓ Pulse: shape and timing control

Two and three terminal devices

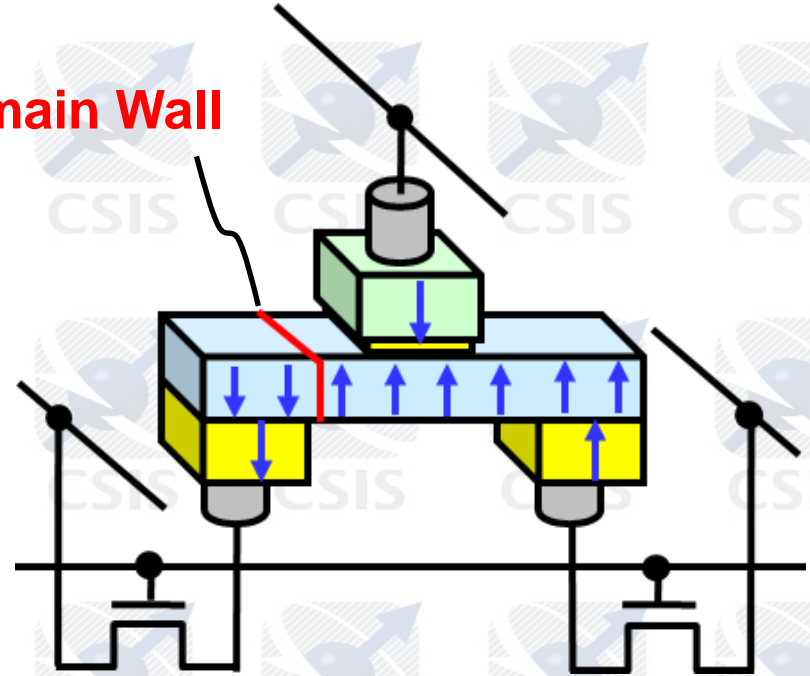


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two terminal

Domain Wall

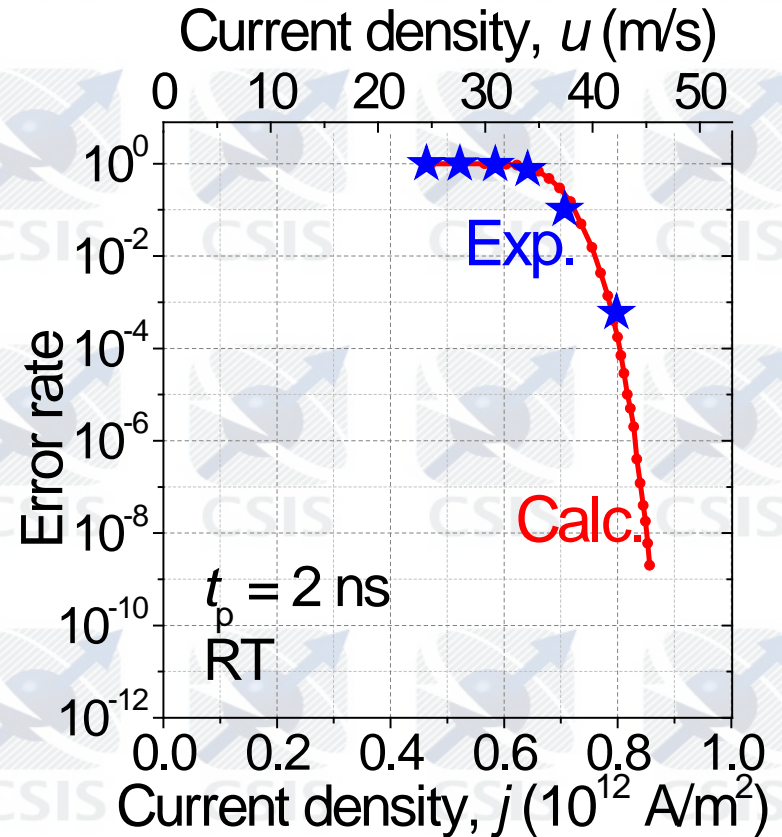
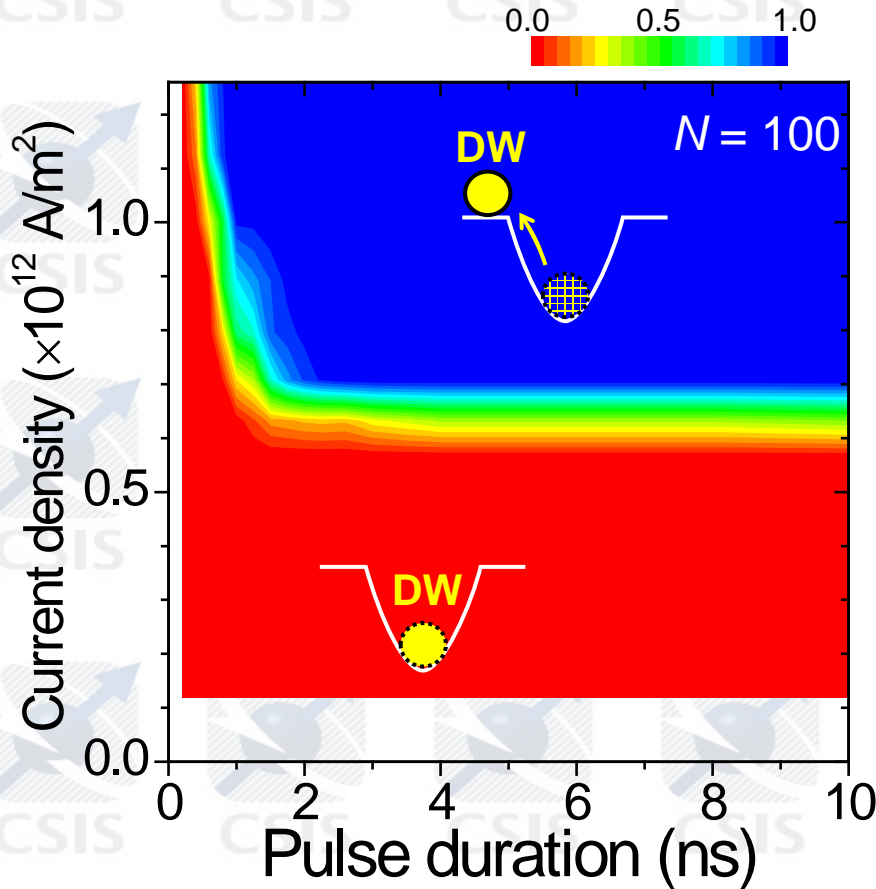


three terminal

Nonvolatile, fast, low voltage and high endurance



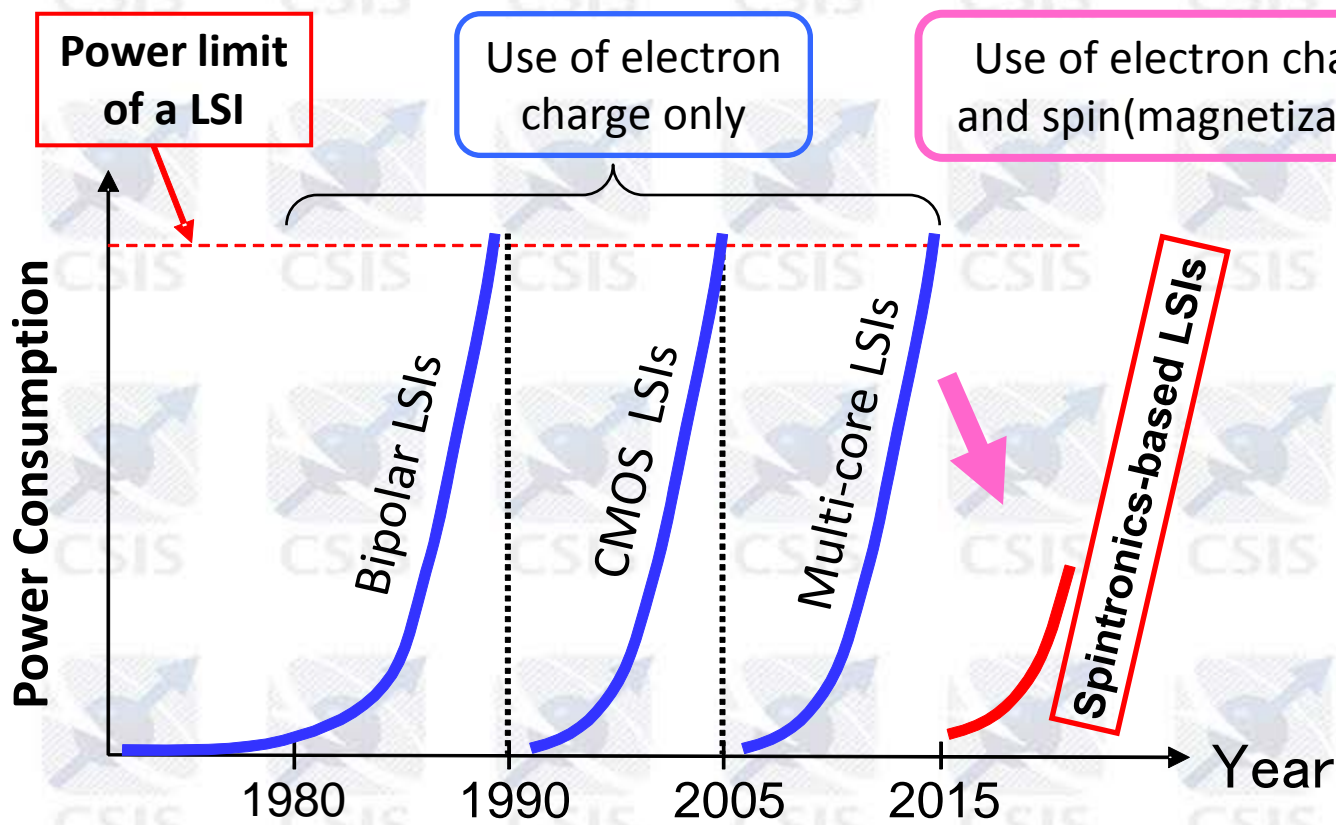
Depinning probability of DW by ns pulses



- Critical current density shows constant down to ~ 2 ns.
- Error rate decreases above a threshold more steeply than MTJs.

A New Paradigm in VLSI

Whenever power consumption of LSI increased to hit a limit of heat dissipation, a paradigm shift in LSI technology has taken place by bringing in new technology.



Summary



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Spintronics devices are an indispensable ingredient in developing CMOS VLSI with low power and high performance.

Two terminal device

- Size dependence of energy barrier of perpendicular CoFeB-MgO MTJ between 30 and 11 nm; size dependence of demagnetization.
- Size dependence of Δ/I_{C0} suggests additional reduction of dissipative path as size reduces.
- Electric-field manipulation of magnetization

Three terminal device

- Depinning probability that determines error rate was explored and shown to follow a function steeper than that known for MTJs.